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# Nonlinear Optical Characterization of CsPbBr<sub>3</sub> Nanocrystals as a Novel Material for the Integration into Electro-Optic Modulators

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Abstract. The present work is concerned with the investigation of the nonlinear optical response of green emissive CsPbBr<sub>3</sub> nanocrystals, in the form of colloidal dispersions in toluene, synthesized via a room-temperature ligand-assisted supersaturation recrystallization (LASR) method. After carrying out a preliminary characterization via X-Ray Diffraction (XRD) and Absorption and Photoluminescence (PL) Spectroscopies, the optical nonlinearity of the asobtained colloids is probed by means of a single-beam Z-scan setup. Results show that the material in question, within the sensitivity of the experimental apparatus, exhibits a nonlinear refractive index n<sub>2</sub> that is the order of  $10^{-15}$  cm<sup>2</sup>/W. Moreover, a three-photon absorption mechanism (3PA) is postulated, according to the fitting of the recorded Z-scan traces and the fundamental absorption threshold, which turns out to be off resonance with twice the energy of the laser radiation. A figure of merit is, then, calculated as an indicator of the quality of the CsPbBr<sub>3</sub> nanocrystals as a candidate material for photonic devices, for instance, Kerr-like electro-optic modulators (EOMs).

## Introduction

The past decade has been characterized by the flourishing development of fiber-integrated circuits and photonic devices [1] based on silicon-organic-hybrid (SOH) slot waveguides [2], namely two silicon rails separated by a submicrometric-wide slot infiltrated by a nonlinear optical (NLO) medium. Polymers and polymer-dye host-guest systems have been largely employed as active materials suitable for exploiting both the linear and quadratic electro-optic (EO) effects [3] [4] [5]. However it is well-known, that organic systems are usually affected by a remarkable two-photon absorption (2PA) that can diminish the effectiveness of the third-order nonlinearity, similarly to the case of silicon [6]. Moreover, the chemical and thermo-mechanical stability of organic materials is often undermined by the conditions at which the device is required to operate, especially at high temperatures. Hence, the quest for novel and promising

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nonlinear optical active media that can meet these requirements is ongoing and lead halide perovskite nanocrystals are part of this framework.

Lead halide perovskites have emerged recently as promising materials for many applications in photovoltaics [7] [8] and optoelectronics [9] [10]. As concerns the field of photonics, latest works have shown some novel opportunities for the integration into nonlinear optical devices [11] [12], thanks to the low-cost fabrication and agile processability of these materials. Moreover, they exhibit remarkable electronic and optical properties such as relatively high values of the refractive index, broadband bandgap tunability, large optical gain and strong nonlinear response, which are enhanced at the nanoscale [13].

### **Experimental Section**

Materials. Chemicals are employed as received, according to their original technical grade: CsBr (99.999%, Sigma-Aldrich), PbBr2 (99.999%, Sigma-Aldrich), DMF ( $\geq$  99.5%, Carl Roth), DMSO ( $\geq$  99.5%, Carl Roth), toluene ( $\geq$  99.5%, Carl Roth), oleylamine ( $\geq$  96%, Fisher Scientific) and oleic acid (90%, Fisher Scientific).

Synthesis of CsPbBr<sub>3</sub> nanocrystals. CsPbBr<sub>3</sub> nanocrystals are synthesized via roomtemperature ligand-assisted supersaturation recrystallization (LASR) [14]. A mixed precursor solution is prepared by dissolving 0.04 mmol CsBr in 0.5 mL DMF and 0.04 mmol PbBr<sub>2</sub> in 0.5 mL DMSO. Alternatively, the dissolution of salts into the aprotic polar solvents is reversed, namely CsBr in DMSO and PbBr<sub>2</sub> in DMF (preparation B). Afterwards, the surface ligands, namely oleylamine (20  $\mu$ L) and oleic acid (10  $\mu$ L), are added to the as-obtained 1 mL precursor solution: they help to control the size of the nanocrystals and to disperse them into the antisolvent. Stirring at 70-75°C for ca. 30 min is accomplished for promoting the complete dissolution of the salts. Hence, 0.25 mL of precursor solution are swiftly injected toluene which acts as the anti-solvent, i.e. a poor solvent for the ions from the precursor salts. The drop between the precursor solubility into the aprotic polar solvents and that into toluene, thus, promotes the formation of a supersaturated state in which the recrystallization of the perovskite phase takes place. Agitation is kept for approximately 1 min, that is the estimated time for the complete formation of nanocrystal [15]. If the vial is irradiated under UV light, a green bright emission is observed, as an indicator for the occurred recrystallization of the perovskite phase.

X-Ray Diffraction Measurements. XRD measurements are performed in reflection mode on a Panalytical Empyrean Diffractometer, using the  $K_{\alpha}$  fluorescence line of a Cu-anode emitting tube. Bragg- Brentano configuration is used as the incident optical pathway (0.25°-0.5°) divergent slits and a solid state hybrid Pix'cel 3D detector, working in 1D linear mode, accomplishes the detection. Samples are studied in the range  $10^{\circ} < 2\theta < 50^{\circ}$ , in the form of thin films drop-cast on glass substrates, with an estimated thickness of few microns.

Spectral Measurements. Absorption spectra are recorded by means of a double beam spectrophotometer (Perkin Elmer Lambda-19), while photoluminescence (PL) measurements are performed via excitation of the 458 nm line of an  $Ar^+$  laser and collected by a compact spectrophotometer (Flame, OceanOptics). Samples are probed in the form of the as-obtained colloids collected inside 1 mm wide glass cuvettes (Hellma® Analytics).

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Z-Scan Measurements. The single-beam Z-scan setup is shown in Figure 1. The laser radiation employed for the analysis is the fundamental wavelength ( $\lambda = 1064$  nm) of a Nd:YAG solid state laser, passively Q-switched via saturable absorber: the nominal pulse duration is  $\tau_p \approx 30$  ps at a repetition rate of  $\nu \approx 10$  Hz. The wavelength chosen for the analysis is assessed to be suitable on account of the fact that the fundamental harmonic of the Nd:YAG is rather adjacent to the telecom wavelengths  $\lambda = 1300$  nm and  $\lambda = 1550$  nm. It seems reasonable to expect that the variation of the NLO coefficients in this wavelength range be small, if compared to that for shorter wavelengths.



*Figure 1*: Schematic of the single-beam Z-scan apparatus. The acquisition is accomplished through an oscilloscope, whose readout is monitored via LabVIEW. A linear controller interfaced with the PC remotely controls the motor stage: this displaces the sample towards the detector  $D_1$ , namely along the positive z-direction.

The output laser radiation passes through a telescope, which acts as a beam expander and is used for improving the beam quality. Consequently, two mirrors ( $M_1$  and  $M_2$ ) deflect the beam onto the Z-scan line, where the beam alignment is controlled by the apertures  $A_2$  and  $A_3$ . The beam splitter (BS) splits the beam in two parts:  $\approx 60\%$  is sent to the probe detector  $D_1$ , while  $\approx 40\%$  to the reference detector  $D_2$ . A converging lens (L) is used for focusing the beam on the sample, while filters (F) are used for attenuating the laser energy and for preventing the saturation of photodetectors, namely, two Si PIN photodiodes with a rise time  $t_R \approx 35$  ns.

#### **Results and Discussion**

Spectral and XRD Characterization. A signature for the formation of green emissive CsPbBr<sub>3</sub> nanocrystals is represented by the emission peak appearing at 520 nm, as reported in Fig. 2. The fluorescence was excited at 458 nm (edge-filter at 480 nm). The absorption threshold occurring **around 500 nm is usually observed in nanocrystals smaller than 50 nm [15] [16] [17].** 



**Figure 2**: Typical absorption and PL spectra of green emissive CsPbBr3 NCs colloids prepared via LASR, showing the characteristic absorption threshold occurring at wavelengths around 500 nm, while a quite strong emission is seen to be peaked at 519 nm. Spectra are reported on the same intensity scale only for qualitative purposes.

The XRD spectra of the perovskite thin films are shown in Fig. 3: both samples present the dominant CsPbBr<sub>3</sub> monoclinic phase (ICDD N0. 00- 018- 0364), characterized by lattice constants a = b = 5.82 Å and c = 5.87 Å, and angles  $\alpha = \beta = 90^{\circ}$  and  $\gamma = 89.65^{\circ}$ . This phase is characterized by the signature peaks at  $2\theta = 15.1^{\circ}$  and  $2\theta = 15.2^{\circ}$ , referred to the diffraction from crystallographic planes (001) and (100) respectively;  $2\theta = 21.5^{\circ}$  and  $2\theta = 21.7^{\circ}$  from planes (110) and (-110);  $2\theta = 30.4^{\circ}$  and  $2\theta = 30.7^{\circ}$ , from planes (002) and (200).



**Figure 3**: XRD spectra of thin films from preparations A and B, along with the labeling (colored squares) of XRD reflections from CsPbBr<sub>3</sub> monoclinic and rhombohedral Cs<sub>4</sub>PbBr<sub>6</sub> phases

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Interestingly, the spectrum referring to preparation B exhibits also fingerprinting peaks of the rhombohedral Cs<sub>4</sub>PbBr<sub>6</sub> phase (ICDD No. 01- 073- 2478) with lattice constants a = b = 13.73 Å and c = 17.32 Å, and angles  $\alpha = \beta = 90^{\circ}$  and  $\gamma = 120^{\circ}$ , even if the monoclinic phase is still predominant. We attribute this to the higher reactive amount of Cs<sup>+</sup> in preparation B, which tends to react with PbBr<sub>2</sub> dissolved in DMF to form Cs<sub>4</sub>PbBr<sub>6</sub>, confirming the results of Yang et al. [16]. However, only the pure CsPbBr<sub>3</sub> NCs colloids obtained from preparation A are devoted to the Z-scan measurements, since, at this point of the analysis, understanding the contribution of the Cs<sub>4</sub>PbBr<sub>6</sub> phase to the NLO activity of the nanocrystals is beyond the scope of this work. The average size of the ordered polycrystalline domains has been estimated by means of the Scherrer formula. Peak analysis yields a mean grain size of  $D_A = 23 \pm 1$  nm and  $D_B = 35 \pm 1$  nm for the CsPbBr<sub>3</sub> monoclinic phase resulting from the two preparations, comparable to those reported in [16] [17].

NLO Characterization via Z-Scan: Closed-Aperture Configuration. CS<sub>2</sub> has been used to calibrate the as-built single-beam Z-Scan setup – as originally developed by Sheik-Bahae et al. [18] - in closed-aperture configuration (CA). CS<sub>2</sub> is a standard material for this type of measurements whose nonlinear refractive index is well-known in the literature [19] [20]:  $n_2 = (3.1 \pm 0.2) \cdot 10^{-14} \text{ cm}^2/\text{W}$ . To the best of the beam quality optimization, laser energy fluctuations are estimated to be in between 10 - 15%: a cuvette filled with CS<sub>2</sub> is, thus, probed for calibrating the irradiance  $I_0$ . Data acquisition is performed by using a 10 cm focal length lens, a diaphragm with aperture radius  $r_a \approx 0.5$  mm and linear transmittance  $S \approx 0.4$  %, which is sufficiently small to consider the limiting case  $S \rightarrow 0$  for the CA peak-valley transmittance  $\Delta T_{CA}$ :

$$\Delta T_{CA} \cong 0.406 |\Delta \Phi_0|. \tag{1}$$

 $\Delta \Phi_0$  is the time-averaged nonlinear phase shift:

$$\Delta \Phi_0 = k \frac{n_2}{\sqrt{2}} I_0 L_{eff} \tag{2}$$

where k is the radiation wavenumber and  $L_{eff}$  the optical path inside the cuvette. A z-resolution of  $\delta z = 0.5$  mm is found to be a good compromise in between the reproducibility of measurements and the irradiation time that samples are subject to, in order to prevent the potential build-up of thermal effects.

The experimental data are fitted with the theoretical curve for the closed-aperture configuration transmittance [16]:

$$T_{CA}(z,\Delta T) = a + 4 \left(\frac{z+b}{z_R}\right) \frac{\Delta T}{0.406} \frac{1}{\left[\left(\frac{z+b}{z_R}\right)^2 + 9\right] \left[\left(\frac{z+b}{z_R}\right)^2 + 1\right]}$$
(3)

where  $\Delta \Phi_0$  is expressed by Eq. (2),  $z_R$  is the beam Rayleigh length while *a* and *b* are scaling factors: the former accounts for the normalized transmittance offset and the latter for the focus position. Average pulse energy and beam diameter are monitored by using a beam profiler (WinCamD): the beam waist is estimated to be  $w_0 = 24 \pm 1 \ \mu m$  at an irradiance of  $I_0 = 14 \pm 2 \ \text{GW/cm}^2$ .

Z-scan measurements on solutions and colloids are affected by the solvent effect: thus, it is necessary to accomplish a scan on the solvent as well, in order to subtract any nonlinear contribution from that of the entire solution. CA scans of the solvent (toluene) and the CsPbBr<sub>3</sub>

NCs colloidal dispersions at three different concentrations (namely 1.5, 2.5 and 4 mM) are shown in Fig. 4.

It must be noticed that when increasing the concentration, scattering processes become predominant: this is apparent from the "humps" emerging in the linear region and, especially, in the asymmetry - with respect to the focus position - affecting the whole Z-scan trace. Fitted  $\Delta T$  and calculated  $n_2$  values for the perovskite NCs are reported in Tab. 1.

The magnitude and the sign of the NLO coefficients usually depend on several factors:

• Radiation-related: peak power and repetition rate, when too high, usually lead to negative nonlinearities and saturation that can be ascribed to the build-up of thermooptic effects. Another contribution arises, obviously, from the operating wavelength due to the dispersion of the NLO coefficients.



• Medium-related: degree of crystallinity, morphology and size.

Figure 4: CA Z-scan traces of CsPbBr<sub>3</sub> NCs colloidal dispersions in toluene at a) 0 mM, b) 1.5 mM, c) 2.5 mM, d) 4 mM.

As a matter of fact, it is usually difficult to compare NLO coefficients obtained from a Z-scan when not taking into account all of the above-mentioned parameters. To the best of our knowledge, although the literature concerning Z-scan measurements of colloidal CsPbBr<sub>3</sub> NCs is still relatively scarce, the n<sub>2</sub> values reported in this work are in good agreement with the results of Liu et al. [21], who found a nonlinear refractive index  $n_2 \approx 0.7 \cdot 10{-}14 \text{ cm}^2/\text{W}$  for green emissive CsPbBr<sub>3</sub> nanocrystals (D = 21.4 nm) at 787 nm.

NLO Characterization via Z-Scan: Open-Aperture Configuration In order to investigate NLA, the aperture  $A_3$  placed in front of the detector  $D_1$  is fully opened (S = 100%). Fig. 5 shows the OA Z-scan traces for the perovskite colloidal dispersions at the same three different concentrations as for CA and for toluene.

Conc. [mM]	ΔΤ	$n_{2,P}[\cdot 10^{-14}  cm^2/W]$
1.5	$0.29\pm0.01$	$0.4 \pm 0.1$
2.5	$0.31\pm0.02$	$0.5\pm0.1$
4	$0.34 \pm 0.02$	$0.6 \pm 0.1$



Figure 5: OA Z-scan traces of CsPbBr3 NCs colloidal dispersions in toluene at a) 0 mM, b) 1.5 mM, c) 2.5 mM, d) 4 mM.

Since the solvent exhibits a negligible nonlinear absorption within the sensitivity of the asbuilt setup, the NLA coefficients returned by the fit, are specifically referred to the perovskite NCs. Experimental data are fitted with the theoretical curves referred to two- and three-photon absorption transmittance [22]:

$$T_{OA}(z,\beta_2) = \sum_{m=0}^{\infty} \frac{(-1)^m}{(m+1)^{\frac{3}{2}}} \left[ \frac{\beta_2 I_0 L_{eff}}{1+z^2/z_R^2} \right]^m \quad ; \tag{4.a}$$

$$T_{OA}(z,\beta_3) = \sum_{m=1}^{\infty} \frac{(-1)^{m-1}}{(2m-1)!(2m-1)^{\frac{1}{2}}} \left[ \frac{\sqrt{2\beta_3 I_0^2 L_{eff}^{(2)}}}{1 + \frac{(z+b)^2}{z_R}} \right]^{2(m-1)}$$
(4.b)

where the effective lengths can be both approximated with the cuvette width at  $\lambda = 1064$  nm, and  $\beta_2$  and  $\beta_3$  represent the 2PA and 3PA coefficients respectively. A three-photon absorption mechanism occurring in perovskite NCs is, thus, postulated, as evidenced from the fact that, within the accuracy guaranteed by the *z* resolution, the 3PA theoretical curve fits the experimental points better than the 2PA curve. This is reasonable when also considering that the wavelength corresponding to twice the laser energy  $\lambda_{2\omega} = 532$  nm falls outside of the absorption band, while the wavelength corresponding to thrice the laser energy  $\lambda_{3\omega} \approx 355$  nm is clearly inside the resonance region.

However, it must be clarified that in order to corroborate this hypothesis, further investigation is required. For instance, a useful tool is represented by a wavelength-dependent Z-scan analysis, in order to probe the dispersion of the multiphoton absorption coefficients, which follow specific scaling rules as a function of the radiation energy [23] [24]. The  $\beta_3$  values returned by the fit are shown in Table 2.

Conc. [mM]	$\beta_{3,P}[\cdot 10^{-2} \text{ cm}^3/\text{GW}^2]$
1.5	$1.2 \pm 0.2$
2.5	$1.4 \pm 0.2$
4	$1.8 \pm 0.2$

*Table 1*:  $\beta_3$  values for perovskite colloidal dispersions at different concentrations.

A further proof of an apparent 3PA process is given by the work of Manzi et al. [25], who found a 2PA-to-3PA transition at about 1030 nm by studying the nonlinear absorption-induced PL in green emissive CsPbBr<sub>3</sub> nanocrystals (D = 10 - 15 nm) by means of a 15ps-pulsed excitation ensured by a tunable laser working in the range 680 – 1080 nm.

## Conclusions

We have presented an assessment of the quality of green emissive CsPbBr<sub>3</sub> nanocrystals as a potential active medium for electro-optic modulators and other photonic devices exploiting the Kerr effect, on the basis of the NLO coefficients yielded by the Z-scan analysis. Besides the remarkable magnitude of the nonlinear refraction, the postulated 3PA mechanism in the mid-IR range is, indeed, preferable – from a probabilistic point of view - to a 2PA process, which is usually observed in organic materials, within the computation of the optical losses inside a waveguide. Furthermore, the dispersion into an optically-inactive polymer matrix can be also considered as a good strategy for both preventing aggregation mechanism among the perovskite nanocrystals and at the same time improving their stability once deposited into the waveguide.

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